

# Synthesis and characterization of layered zinc hydroxychlorides

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## Abstract

Layered material of zinc hydroxychlorides ( $\text{Zn}_5(\text{OH})_8\text{Cl}_2 \cdot n\text{H}_2\text{O}$ : ZHC), which is one of the basic zinc salts (BZS), was synthesized from ZnO nano-particles aged with aqueous  $\text{ZnCl}_2$  solutions at different temperatures ranging from 6 to 140 °C for 48 h. X-ray diffraction (XRD) results indicated that the diffraction peaks of ZnO completely disappeared by aging at 6 °C and the ZHC peaks were developed. By increasing the aging temperature, crystallinity of the layered structure was improved. At 6 °C, the ZHC particles were thin hexagonal plate particles with sizes ranging from 1 to 3  $\mu\text{m}$ . The particle size of ZHC was independent of aging temperature. The atomic Cl/Zn ratios of all the ZHC materials were almost 0.2 less than 0.4 of the theoretical ratio, indicating that the synthetic ZHC is Cl-deficient. It seemed that half of Cl atoms in the layer were replaced with  $\text{HCO}_3^-$  and/or  $\text{OH}^-$ . The specific surface areas of ZHC estimated from  $\text{N}_2$  adsorption isotherms were ca.  $10\text{ m}^2\text{ g}^{-1}$  and were independent of the aging temperature. However, the  $\text{H}_2\text{O}$  monolayer adsorption capacity per unit surface area ( $n_w$ ) for all the samples was higher than that of ZnO particles, revealing the high affinity of ZHC to  $\text{H}_2\text{O}$  molecules. The  $n_w$  values were increased by reducing the crystallinity of ZHC. This enhancement of  $\text{H}_2\text{O}$  adsorption selectivity was thought to be related with less-crystallized parts of the particles.

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**Keywords:** Zinc hydroxychloride; Layered material; Crystal structure; Gas adsorption; Adsorption selectivity

## 1. Introduction

Basic zinc salts of (BZS), e.g., zinc hydroxychlorides ( $\text{Zn}_5(\text{OH})_8\text{Cl}_2 \cdot \text{H}_2\text{O}$ : simonkolleite or ZHC), zinc hydroxycarbonate ( $\text{Zn}_5(\text{CO}_3)_2(\text{OH})_6$ : hydrozincite) and zinc hydroxysulfate ( $\text{Zn}_4\text{SO}_4(\text{OH})_6 \cdot n\text{H}_2\text{O}$ ), possess a layered structure [1]. Similar materials of layered double hydroxide ( $[(M_{1-x}^{2+}M_x^{3+}(\text{OH})_2)_x]A_{n/x}^{n-} \cdot m\text{H}_2\text{O}$ , where  $M^{2+}$  and  $M^{3+}$  are divalent and trivalent metal ions, respectively: LDH) and hydroxy double salts ( $[(M_{1-x}^{2+}M'^{2+})(\text{OH})_{3(1-y)}]^{+}A_{(1+3y)/n}^{n-} \cdot m\text{H}_2\text{O}$ , where  $M$  and  $M'$  are divalent metal ions such as  $\text{Zn}^{2+}$ : HDS) have been synthesized by many researchers and the materials have received attention in various applications such as catalysts, anion exchangers, adsorbents and so on [2–12]. However, synthesis and nature of BZS, particularly ZHC, have not been fully elucidated. Since the ZHC possesses the basal nano-sheets

mainly composed of Zn(II) octahedron and tetrahedron and the  $\text{Cl}^-$  and  $\text{H}_2\text{O}$  molecules exist in the layer [1,13–16], the material is anticipated to be available for anion exchangers, adsorbents for gases, etc. However, to our knowledge, no study has been reported on the synthesis and nature of ZHC except for the study by Ishikawa et al. [17,18]. Recently, Ishikawa et al. [17,18] reported the synthesis of ZHC from aqueous  $\text{ZnCl}_2$  solutions in the presence of Al(III) ions and indicated that the increasing atomic Al/Zn ratio in the starting solutions varied the crystal phases of the products as  $\text{ZnO} \rightarrow \text{ZnO} + \text{ZHC} \rightarrow \text{ZHC} \rightarrow \text{LDH}$ . They further concluded that ZHC formation is effectively promoted by the addition of Al(III) ions. Nonetheless, Zn(II) atoms in the synthetic ZHC were partly replaced with Al(III). To elucidate the detailed information about formation condition and characterization of ZHC, synthesis of ZHC freed from Al(III) is required.

To clarify the formation, the condition of ZHC is also important in corrosion science. In industry, steels are often

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galvanized by different zinc alloys such as Zn–Al, Zn–Al–Mg, Zn–Al–Mg–Si to suppress the corrosion of steels. The high corrosion resistance was explained by the fact that the protecting property of zinc hydroxycarbonate formed in the solution containing  $\text{Cl}^-$  is related to the formation of a disordered layer structure producing adherent films [19]. Further, it has been reported that the zinc rusts contain various kinds of oxides, hydroxides and BZS depending on the exposure environment [1,19–24] and the ZHC is formed together with ZnO by exposing the Zn-coated steels to the industrial or marine atmosphere [20].

The aim of this study was to clarify the detailed nature of the ZHC, hence we tried to synthesize the ZHC by aging ZnO nano-particles in aqueous  $\text{ZnCl}_2$  solutions at different temperatures of 6–140 °C and  $\text{ZnCl}_2$  concentrations. The products thus obtained were characterized by X-ray diffraction (XRD), transmission electron microscopy (TEM), thermal analysis, inductively coupled plasma emission spectroscopy (ICP-AES), gas adsorption measurements and so on. The formation conditions and functions of ZHC are discussed in this article.

## 2. Experimental

### 2.1. Synthesis of ZnO particles

ZnO particles were synthesized by the following wet method. An aqueous solution ( $1.0 \text{ dm}^3$ ) of  $50.0 \text{ mmol dm}^{-3}$   $\text{ZnCl}_2$  was prepared and the pH of the solution was adjusted to 9.0 by adding a 28%  $\text{NH}_3$  solution under stirring condition. The resultants were aged at 50 °C for 24 h in a sealed polyethylene terephthalate bottle without stirring. After aging, the resulting precipitates were filtered off, thoroughly washed with deionized-distilled water and finally dried at 50 °C in an air oven for 24 h.

### 2.2. Synthesis of ZHC particles

Preparation of ZHC particles was carried out in aqueous media as follows. The synthetic ZnO particles ( $16.0 \text{ mmol}$ ) were added into 50 ml of  $0\text{--}2.0 \text{ mol dm}^{-3}$   $\text{ZnCl}_2$  solutions in a sealed polypropylene vessel or a stainless-steel autoclave and they were aged at 6–140 °C for 48 h without stirring. The precipitates generated were filtered off, washed with acetone and finally dried in vacuo at 50 °C for 24 h.

### 2.3. Characterization

The materials thus obtained were characterized by a variety of conventional techniques. Powder XRD patterns were taken on a Rigaku diffractometer with a Ni-filtered  $\text{CuK}\alpha$  radiation (30 kV, 16 mA). Particle morphology was observed by a JEOL TEM. Transmission IR spectra were recorded with a resolution of  $4 \text{ cm}^{-1}$  using a KBr method by a JASCO Fourier transform infrared (FTIR) spectrometer. Zn and Cl contents in the materials were respectively assayed by a Seiko ICP-AES and the Mohr method.

The samples were dissolved in  $\text{HNO}_3$  solutions. Thermal gravimetry (TG) and differential thermal analysis (DTA) curves were traced on a Rigaku thermoanalyzer at a heating rate of  $10 \text{ }^\circ\text{C min}^{-1}$  under air stream. Specific surface area was obtained by fitting the BET equation to the  $\text{N}_2$  adsorption isotherms measured by a Quntachrome volumetric apparatus at the boiling temperature of liquid nitrogen. Adsorption isotherms of  $\text{H}_2\text{O}$  were determined by a gravimetric technique at 25.0 °C. Prior to the adsorption measurements, the samples were treated at 100 °C under  $10^{-5}$  Torr for 2 h.

## 3. Results and discussion

### 3.1. Influence of aging temperature and $\text{ZnCl}_2$ concentration on the formation of ZHC

The ZnO particles were aged in  $2.0 \text{ mol dm}^{-3}$  aqueous  $\text{ZnCl}_2$  solutions at different temperatures of 6–140 °C for 48 h. Fig. 1 shows the XRD patterns of the products. Hereafter,  $\text{ZnCl}_2$  concentration is abbreviated as  $[\text{Zn}^{2+}]$ . Pattern a is synthetic ZnO particles used as a raw material of ZHC (JCPDS no. 46-1451). At an aging temperature of 6 °C, the diffraction peaks due to ZnO completely vanish and new peaks mainly develop at  $2\theta = 11.2^\circ$ ,  $22.5^\circ$  and  $46.0^\circ$ , which can be ascribed to the ZHC (no. 7-195). These facts allow us to infer that the ZnO particles are hydrolyzed in  $\text{ZnCl}_2$  solutions to recrystallize as ZHC by the following reactions:

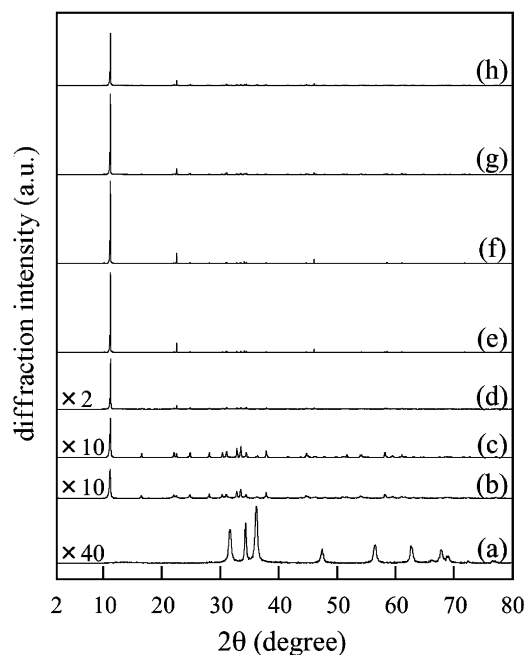
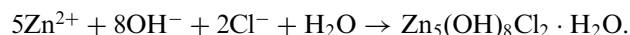
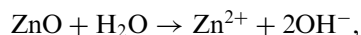


Fig. 1. XRD patterns of: (a) synthetic ZnO and (b–h) ZnO aged at  $[\text{Zn}^{2+}] = 2.0 \text{ mol dm}^{-3}$  and different temperatures for 48 h. Aging temperature (°C), (b) 6, (c) 30, (d) 50, (e) 85, (f) 100, (g) 120 and (h) 140.

The diffraction intensity of ZHC, particularly the peak at  $2\theta = 11.2^\circ$ , is increased by raising the aging temperature. Further, no remarkable differences in position and shape of the peaks are recognized. Fig. 2 plots the diffraction intensity of (110) and (003) peaks, which respectively correspond to the crystallinity of *a*- or *b*-direction and *c*-direction of the crystal, as a function of aging temperature. It is noteworthy that the diffraction intensity of the (003) peak dramatically increases as the aging temperature rises and is maximum at  $120^\circ\text{C}$ . However, the intensity of the (110) peak is independent of the aging temperature. These facts prove that the increment of aging temperature improves the crystallinity of the layered structure. The interlayer *d* distance of the products estimated from the (003) diffraction peak ranges from 0.787 to 0.789 nm. On the other hand, the unit cell dimension *a*

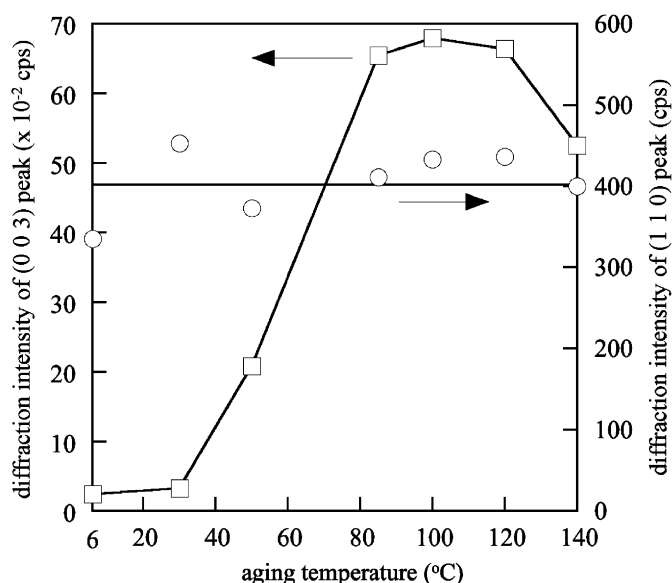


Fig. 2. Plots of diffraction intensity of: ( $\square$ ) (003), and ( $\circ$ ) (110) peaks of ZHC as a function of aging temperature.

calculated from the (110) peak is 0.634–0.635 nm. The *d* and *a* values obtained are comparable to the literature values (*d* = 0.788 nm, *a* = 0.634 nm) [1]. Unfortunately, the crystallite size of the materials could not be determined because the values are beyond the evaluation limit of the crystallite size by the Scherrer equation ( $\sim 300$  nm).

Fig. 3 displays the TEM pictures of the products formed at different aging temperatures and  $[\text{Zn}^{2+}] = 2.0 \text{ mol dm}^{-3}$ . The ZnO synthesized is fine irregular particles (picture a). At  $6^\circ\text{C}$ , thin hexagonal plate particles with the sizes of 1–3  $\mu\text{m}$  are seen and the fine ZnO particles completely disappear (picture b). As confirmed by XRD measurements, the ZHC is formed at the same aging temperature. It is, therefore, indicated that the hexagonal plate particles are identified as ZHC. No remarkable change in particle size of ZHC is recognized with an increase of aging temperature (pictures c and d).

The Zn and Cl contents of the products were almost constant at 9.32–9.54 and 1.80–1.93  $\text{mmol g}^{-1}$  over the whole aging temperature, respectively. The theoretical Zn and Cl contents calculated from the chemical formula of ZHC ( $\text{Zn}_5(\text{OH})_8\text{Cl}_2 \cdot \text{H}_2\text{O}$ ) are 9.06 and 3.62  $\text{mmol g}^{-1}$ , respectively. The Zn amount is close to the theoretical value, while the Cl content is almost half of the value, testifying that the materials are Cl-deficient. The starting  $\text{ZnCl}_2$  solutions contain much larger amounts of  $\text{Cl}^-$  than the products. This suggests that the insufficient Cl source does not cause the Cl deficiency. To verify this, IR spectra were traced. Although the results are not shown here, the spectra of all the synthetic ZHC samples exhibited a weak stretching vibration C–O absorption band at  $1410 \text{ cm}^{-1}$  and the area intensity of this band was independent of the aging temperature. The ZHC crystals have a layered structure and the basal sheets were octahedrally and tetrahedrally coordinated with OH and Cl [1]. The Cl atoms are located in each top corner of the Zn(II) tetrahedron and the  $\text{H}_2\text{O}$  molecules are located between

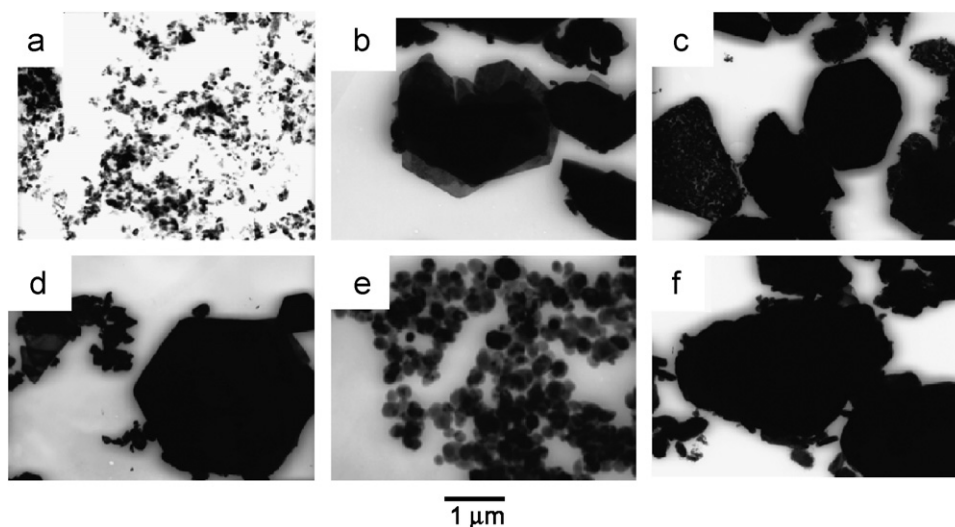


Fig. 3. TEM pictures of: (a) synthetic ZnO, (b–d) ZnO aged at  $[\text{Zn}^{2+}] = 2.0 \text{ mol dm}^{-3}$  and elevated temperatures for 48 h and (e, f) ZnO aged at different  $[\text{Zn}^{2+}]$  and  $140^\circ\text{C}$  for 48 h. Aging temperature ( $^\circ\text{C}$ ), (b) 6, (c) 85, (d) 120.  $[\text{Zn}^{2+}]$  ( $\text{mol dm}^{-3}$ ), (e) 0.2 and (f) 0.5.

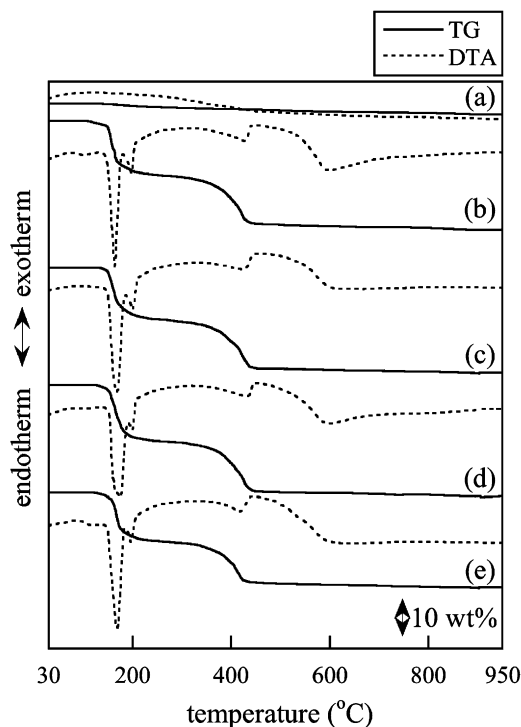


Fig. 4. TG-DTA curves in air of: (a) ZnO and (b–e) ZHC formed at  $[\text{Zn}^{2+}] = 2.0 \text{ mol dm}^{-3}$  and different aging temperatures for 48 h. The solid and dotted lines represented TG and DTA curves, respectively. Aging temperature ( $^{\circ}\text{C}$ ), (b) 6, (c) 30, (d) 100, (e) 120.

the layers bound by weak hydrogen bonds of  $\text{OH}\cdots\text{Cl}$  [1]. Moreover, Ghose has reported that the Cl atoms in the layer are replaced easily with carbonate ions [16]. Besides, Ishikawa et al. [17] recently indicated the replacement of Cl atoms in ZHC with  $\text{HCO}_3^-$ . It seems, therefore, that half of Cl atoms in the layer are replaced with  $\text{HCO}_3^-$  and/or  $\text{OH}^-$ , resulting in Cl deficiency in ZHC.

The thermal stability of ZHC was examined by TG and DTA. Fig. 4 shows the TG and DTA curves taken in an air stream for the products formed at various aging temperatures represented by solid and dotted lines, respectively. The TG curve of ZnO shows a continuous mass loss from 200 to  $950^{\circ}\text{C}$  with no DTA peak, in which the mass loss is mainly due to the removal of  $\text{H}_2\text{O}$  molecules strongly adsorbed and/or involved in the material. The TG curves of the materials obtained at 6 and  $30^{\circ}\text{C}$  undergo a steep mass loss between 150 and  $220^{\circ}\text{C}$  accompanying a strong endothermic DTA peak (curves b and c). As the aging temperature is raised, no remarkable change in mass loss value is recognized, while the endothermic peak slightly shifts to a higher temperature from 165 to  $180^{\circ}\text{C}$ . As stated above, elevating the aging temperature increases the crystallinity of ZHC. It seems that the ZHC yielded at 6– $30^{\circ}\text{C}$  contains poorly crystallized and hydrated parts such as an amorphous phase. Therefore, the mass loss would be due not only to removal of  $\text{H}_2\text{O}$  in the ZHC layer but also to dehydration of the amorphous phase, giving rise to a low-temperature shift of the DTA peak. Furthermore, the wide-range mass loss from 380 to  $500^{\circ}\text{C}$  with an

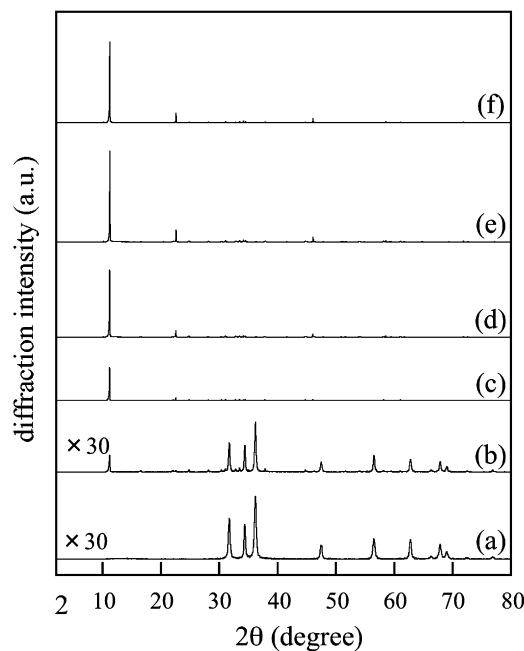


Fig. 5. XRD patterns of: (a) ZnO and (b–f) ZnO aged at different  $[\text{Zn}^{2+}]$  and  $140^{\circ}\text{C}$  for 48 h.  $[\text{Zn}^{2+}]$  ( $\text{mol dm}^{-3}$ ), (b) 0.2, (c) 0.5, (d) 1.0, (e) 1.5 and (f) 2.0.

endothermic peak can be ascribed to the release of OH groups in the materials as  $\text{H}_2\text{O}$  molecules [17].

To verify the influence of  $[\text{Zn}^{2+}]$  in the starting solution on the formation of ZHC, ZnO particles were aged at various  $[\text{Zn}^{2+}]$  and  $140^{\circ}\text{C}$  for 48 h. Fig. 5 shows the XRD patterns of the products. At  $[\text{Zn}^{2+}] = 0.2 \text{ mol dm}^{-3}$ , the diffraction peaks due to ZnO are faintly weakened and the weak peaks of ZHC appear. The ZnO peaks completely disappear at  $[\text{Zn}^{2+}] = 0.5 \text{ mol dm}^{-3}$ . The ZHC peaks are intensified with the increment of  $[\text{Zn}^{2+}]$ , showing a maximum at  $[\text{Zn}^{2+}] = 1.5 \text{ mol dm}^{-3}$ . The morphology of the particles was confirmed by TEM observation and the results are displayed in Figs. 3(e) and (f). When aged at  $[\text{Zn}^{2+}] = 0.2 \text{ mol dm}^{-3}$ , thin hexagonal plate ZHC particles with sizes ranging from 10 to 50 nm are observed in addition to the irregular ZnO particles (picture e). At  $[\text{Zn}^{2+}] = 0.5 \text{ mol dm}^{-3}$ , the ZHC particles grow and the irregular particles diminish, in accordance with the XRD results mentioned above (picture f). No remarkable difference in morphology of the ZHC particles yielded at  $[\text{Zn}^{2+}] \geq 0.5 \text{ mol dm}^{-3}$  was recognized. The atomic Cl/Zn ratios of ZHC produced at  $[\text{Zn}^{2+}] = 0.5\text{--}2.0 \text{ mol dm}^{-3}$  were ca. 0.2 less than 0.4 of the theoretical value, indicating that the materials are Cl-deficient. As suggested by the FTIR result, the Cl deficiency was caused by replacement of one-half of Cl in the layer with  $\text{HCO}_3^-$  and/or  $\text{OH}^-$ .

### 3.2. Adsorption of $\text{N}_2$ and $\text{H}_2\text{O}$

The adsorption property of the synthetic ZHC was examined by the adsorption of  $\text{N}_2$  and  $\text{H}_2\text{O}$ . Fig. 6 shows the adsorption isotherms of  $\text{N}_2$  on ZnO and ZHC



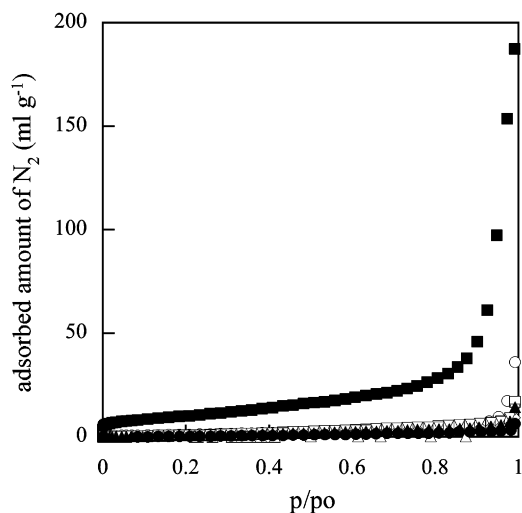


Fig. 6.  $N_2$  adsorption isotherms on (■) ZnO and ZnO aged at  $[Zn^{2+}] = 2.0 \text{ mol dm}^{-3}$  and different temperatures for 48 h. Aging temperature ( $^{\circ}\text{C}$ ), (○) 6, (□) 30, (△) 50, (◇) 85, (▽) 100, (●) 120 and (▲) 140.

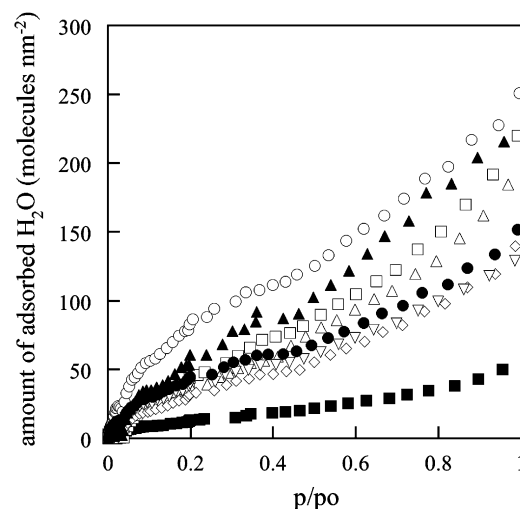


Fig. 7.  $H_2O$  adsorption isotherms on (■) ZnO and ZnO aged at  $[Zn^{2+}] = 2.0 \text{ mol dm}^{-3}$  and different temperatures for 48 h. Aging temperature ( $^{\circ}\text{C}$ ), (○) 6, (□) 30, (△) 50, (◇) 85, (▽) 100, (●) 120 and (▲) 140.

synthesized at different aging temperatures of 6–140  $^{\circ}\text{C}$  for 48 h. The products at  $[Zn^{2+}] = 2.0 \text{ mol dm}^{-3}$  were evacuated at 100  $^{\circ}\text{C}$  for 2 h prior to the adsorption measurements. The adsorption isotherm on ZnO represented by filled squares belongs to type II in the IUPAC classification. The specific surface area ( $S_n$ ) of the material estimated from the BET equation was  $39 \text{ m}^2 \text{ g}^{-1}$ . The adsorbed amounts suddenly decrease with the formation of ZHC and the  $S_n$  values become ca.  $10 \text{ m}^2 \text{ g}^{-1}$ . The decrease of adsorbed amount and  $S_n$  is inferred to be due to the formation of larger ZHC particles than the ZnO ones as confirmed in Fig. 3. No micropores were ascertained by the  $t$ -plot analysis [25] for all the isotherms, meaning that the  $N_2$  molecules cannot access the ZHC layer.

Fig. 7 compares the adsorption isotherms of  $H_2O$  on ZHC formed at various aging temperatures of 6–140  $^{\circ}\text{C}$ . The  $[Zn^{2+}]$  and pretreatment temperature of the adsorption were the same as those of  $N_2$  adsorption. Contrary to  $N_2$  adsorption, the adsorbed amounts of  $H_2O$  on ZHC are larger than that on ZnO. This means that the materials exhibit a higher affinity to  $H_2O$  molecules than the  $N_2$  ones. The adsorbed amount at  $p/p_0 = 1.0$  is maximum at the aging temperature of 6  $^{\circ}\text{C}$  and decreases with increasing the aging temperature, while it slightly increases again at 140  $^{\circ}\text{C}$ . To clarify the difference among these isotherms, monolayer adsorption capacity of  $H_2O$  molecules per unit surface area ( $n_w$ ) based on  $S_n$  is plotted against the aging temperature in Fig. 8. The  $n_w$  value (12.7) of ZnO is close to  $9.3 \text{ mol nm}^{-2}$  of the theoretical  $n_w$  estimated using the  $0.108 \text{ nm}^2$  as the cross-sectional area of a  $H_2O$  molecule, depicted by a dashed line. The  $n_w$  value is maximum at the aging temperature of 6  $^{\circ}\text{C}$  and is decreased with elevating the aging temperature. Fig. 8 plots the intensity of the (003) peak in XRD patterns by open circles as a function of the aging temperature. Clearly, we see that the increase of the diffraction intensity lowers the  $n_w$  value. This fact allows us

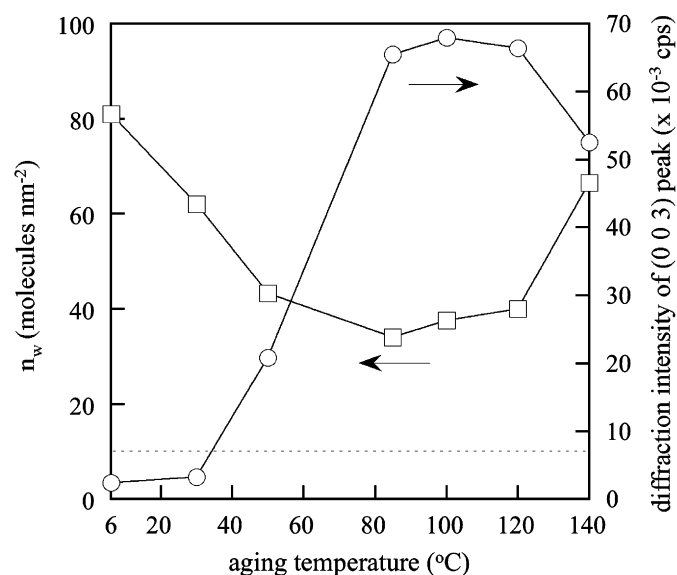


Fig. 8. Plots of (□)  $n_w$  and (○) diffraction intensity of (003) peak of ZHC against aging temperature.

to infer that the selective adsorption of  $H_2O$  molecules is dependent on the crystallinity of ZHC. The crystallinity of ZHC is improved with elevating the aging temperature. Hence, the enhancement of  $H_2O$  adsorption on the products formed at lower aging temperatures of 6–30  $^{\circ}\text{C}$  is thought to be due to the less crystallized and hydrated parts of the particles as confirmed by thermal analysis. It seems that the less-crystallized parts are dehydrated during the pretreatment of adsorption and the rehydration takes place by the  $H_2O$  adsorption. On the other hand, the  $H_2O$  adsorption selectivity falls with increasing crystallinity of ZHC because the amount of less crystallized and hydrated parts is reduced.

#### 4. Conclusions

From the information presented in this publication, we can draw the following conclusions. Layered ZHC can be synthesized from nano-sized ZnO particles aged in aqueous  $\text{ZnCl}_2$  solutions at 6–140 °C for 48 h. Elevating the aging temperature increases the crystallinity of ZHC. The ZHC was hexagonal plate particles of which the size was almost unchanged by raising the aging temperature. The synthetic ZHC was Cl-deficient, which would be caused by exchange of one-half of Cl in the layer with  $\text{HCO}_3^-$  and/or  $\text{OH}^-$ . The  $S_n$  values of ZHC were ca.  $10 \text{ m}^2 \text{ g}^{-1}$  and were independent of the aging temperature. The  $n_w$  value of ZHC, that is, a measure of selective adsorption of  $\text{H}_2\text{O}$ , was higher than that of ZnO, showing the high affinity of ZHC to  $\text{H}_2\text{O}$  molecules. The  $n_w$  value was increased by decreasing the crystallinity of ZHC.

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